

Hydrogen storage in metal hydrides – fundamental principles meet practical life



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C-based economy -> CO₂, Global Warming, dirt
H-based economy -> H₂O – clean, but several challenges

Interaction with hydrogen, H₂ storage

Volume comparison for storage



4kg hydrogen in/as Mg₂NiH₄ LaNi₅H₆ liquid H₂ H₂ at 100atm

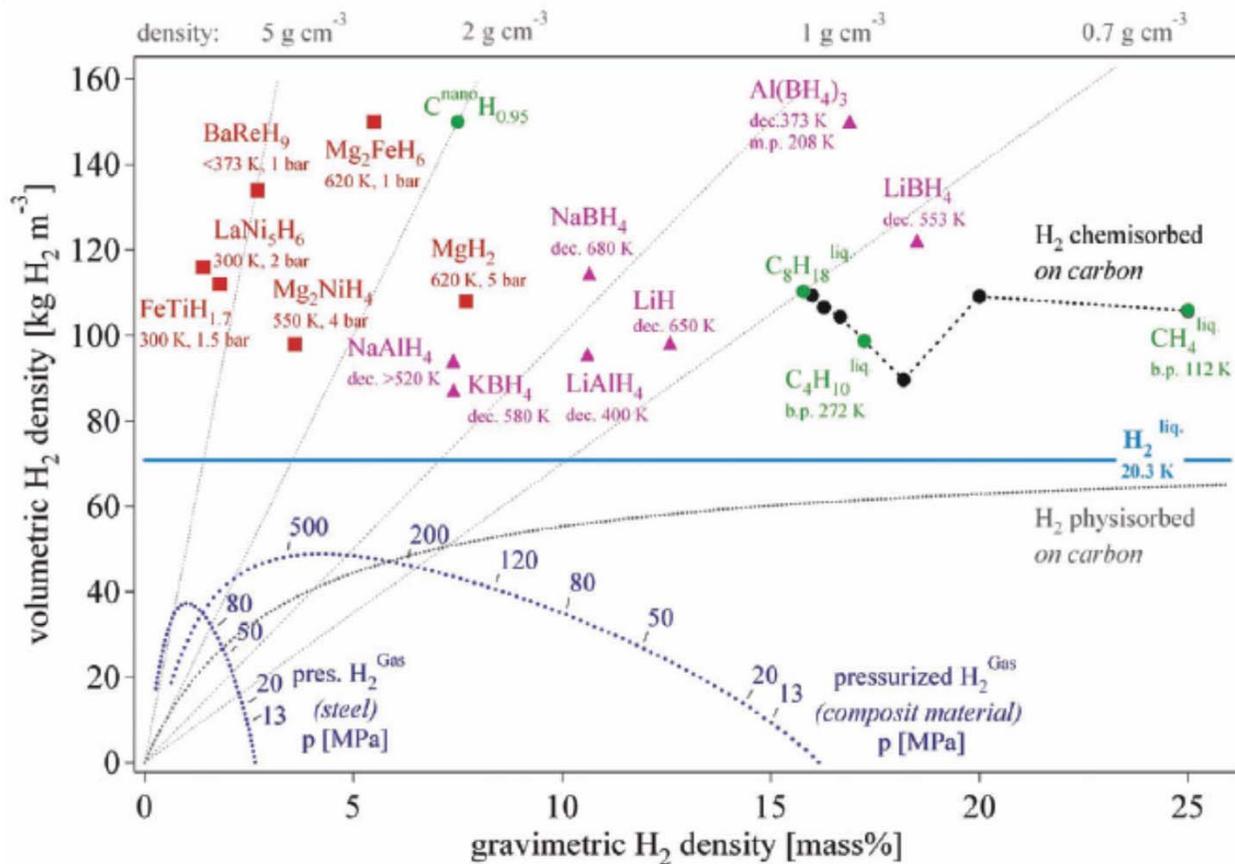
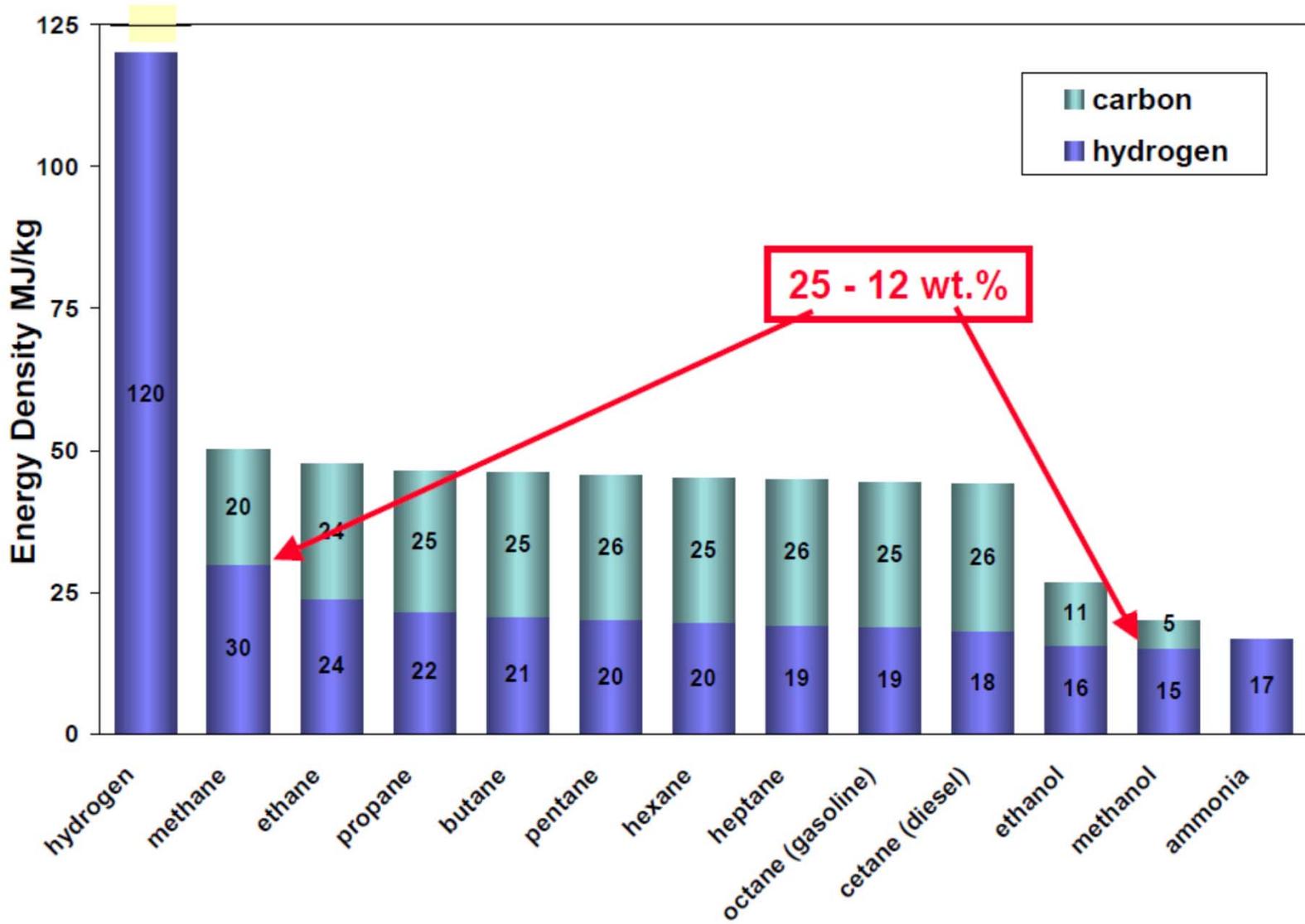


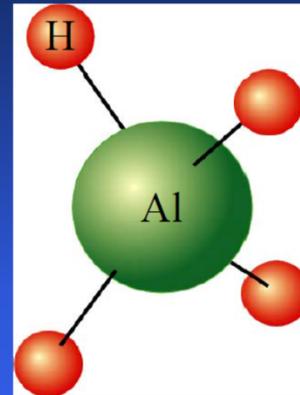
Fig. 7 Volumetric and gravimetric hydrogen density of some selected hydrides. Mg_2FeH_6 shows the highest known volumetric hydrogen density of $150 \text{ kg}\cdot\text{m}^{-3}$, which is more than double that of liquid hydrogen. $BaReH_9$ has the largest H/M ratio of 4.5, i.e. 4.5 hydrogen atoms per metal atom. $LiBH_4$ exhibits the highest gravimetric hydrogen density of 18 mass%. Pressurized gas storage is shown for steel (tensile strength $\sigma_v = 460 \text{ MPa}$, density $6500 \text{ kg}\cdot\text{m}^{-3}$) and a hypothetical composite material ($\sigma_v = 1500 \text{ MPa}$, density $3000 \text{ kg}\cdot\text{m}^{-3}$).

Specific energy of fuels



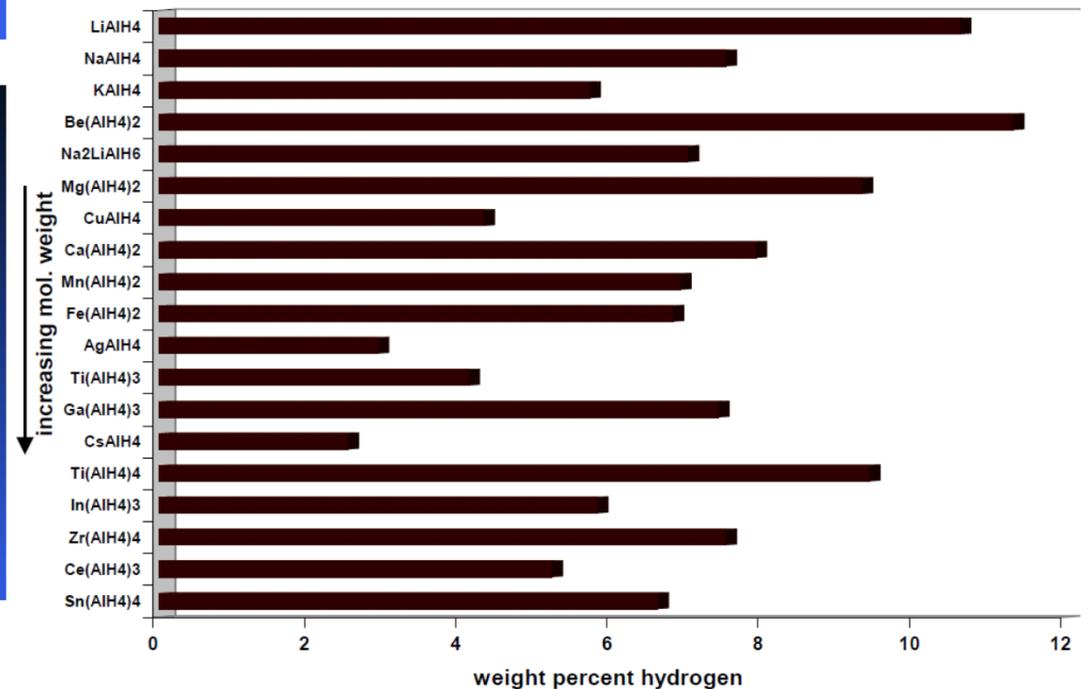
Renewed interest in complex hydrides

- ⇒ Complex hydrides consist of a H=M complex with additional bonding element(s)
- ⇒ Reversibility demonstrated in NaAlH_4
By Bogdanovic and Schwickardi (1996)
- ⇒ Hydrogen complexes include
 - $(\text{AlH}_4)^-$ (alanates)
 - $(\text{BH}_4)^-$ H with Group VIII elements
- ⇒ Advantages:
 - Can have lower formation energy
 - Can have high H/M.



Issues with complex hydrides

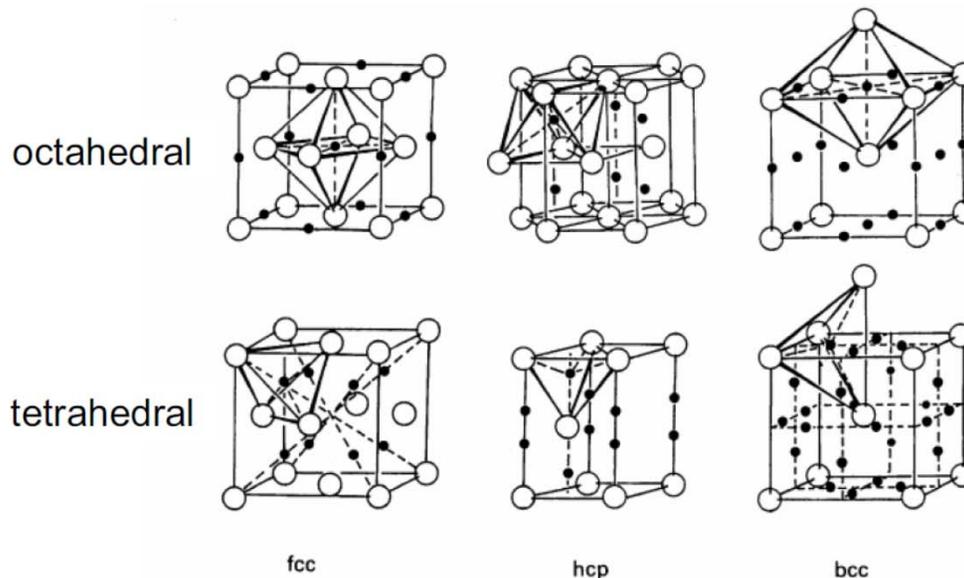
- ⇒ Reversibility
 - Role of catalyst or dopant.
- ⇒ Thermodynamics
 - Pressure, temperature.
- ⇒ Kinetics
 - Long-range transport of heavy species
- ⇒ Capacity



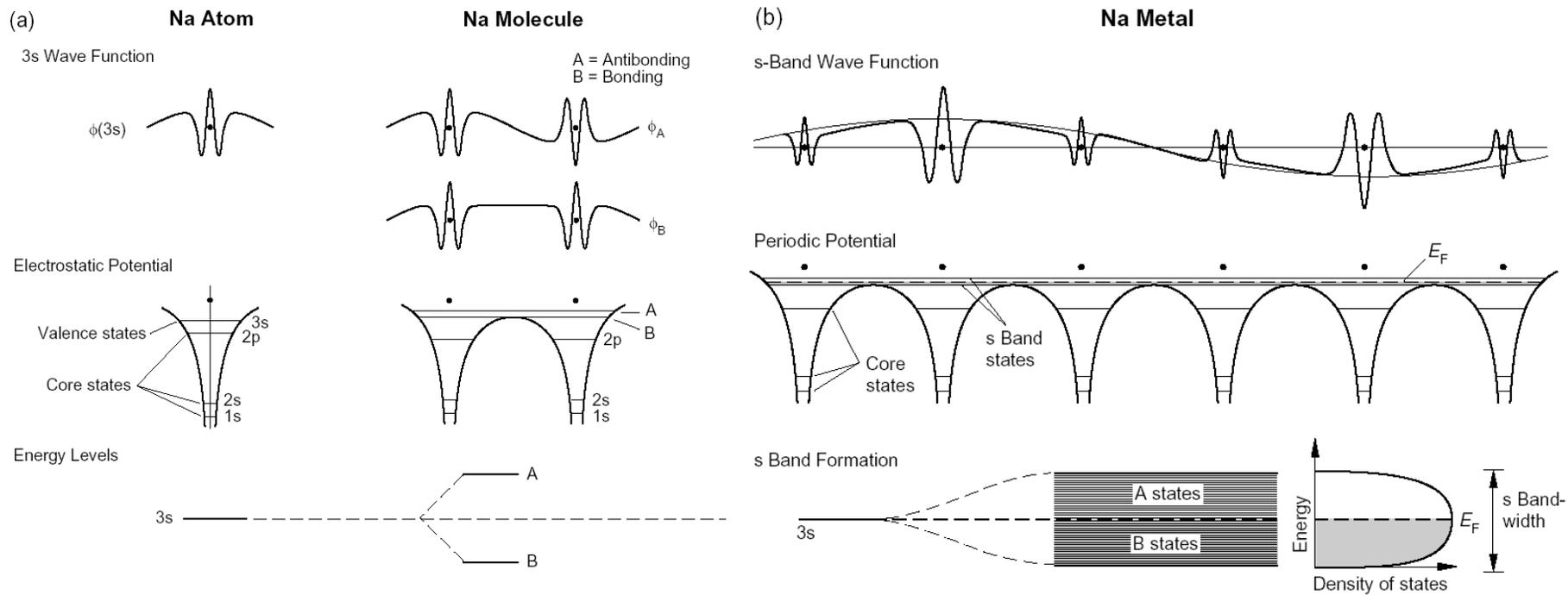
Metal hydrides – not all metals absorb H

Decisive is energy balance...

1. H bonding decreases energy
2. Volume expansion increase energy – work against elastic forces ...several % to 60 % - powder
3. Disruption or weakening of existing bonds increases energy
4. Also dissociation of H₂ molecule plays a role



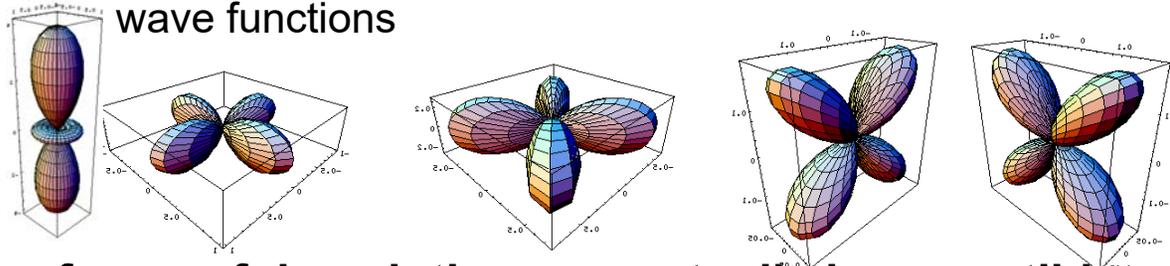
*What are impacts on electrons?
2. Volume expansion
1. H bonding*



Periodic potential – Bloch wave functions, itinerant states

$$\Psi_k(\mathbf{r}) = e^{i\mathbf{k}\cdot\mathbf{r}} u_k(\mathbf{r})$$

But inner shells retain the localized, atomic wave functions



Two forms of description are mutually incompatible – how about the 5f states?

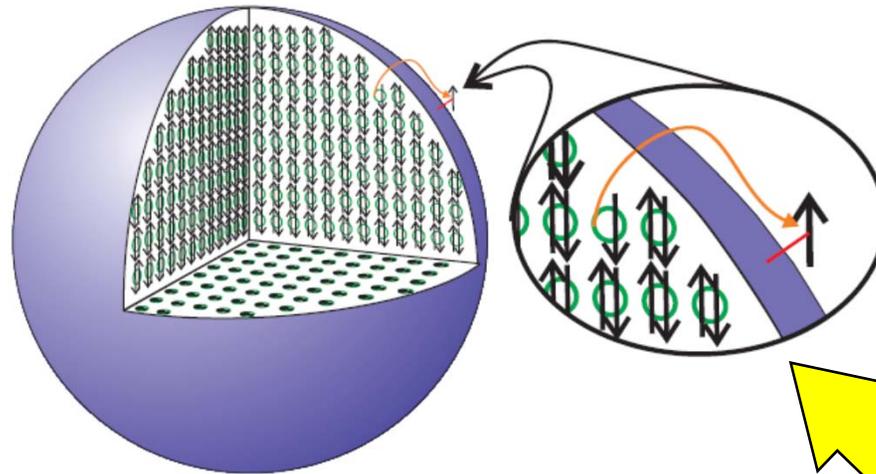
The most interesting phenomena are at the crossover (verge of localization) – qualitatively new cooperative phenomena

Fermi gas – not too bad approximation

Free (non-interacting) electrons + Fermi-Dirac statistics

A. J. Schofield

2



$$E = \hbar^2 k^2 / 2m_e$$

$$\mathbf{p} = \hbar \mathbf{k}$$

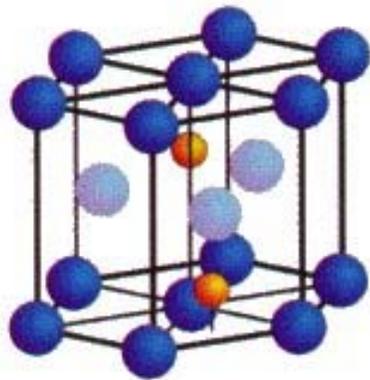
Free electrons in k-space

Figure 1: The ground state of the free Fermi gas in momentum space. All the states below the Fermi surface are filled with both a spin-up and a spin-down electron. A particle-hole excitation is made by promoting an electron from a state below the Fermi surface to an empty one above it.

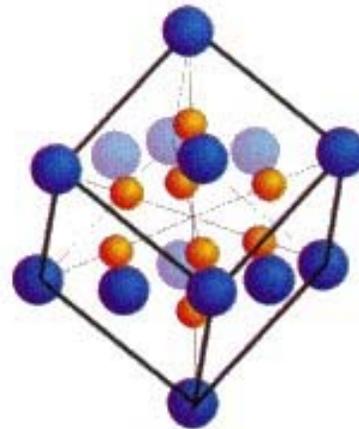
Only small amount of electrons (in the range $k_B T$ around the Fermi surface) affect properties as specific heat, susceptibility, resistivity...

switchable mirrors

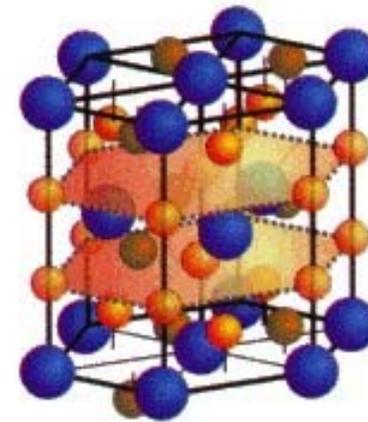
Strongly electropositive **Y** has $1 \cdot 4d$ electron + $2 \cdot 5s$ electrons



YH_0



YH_2



YH_3

hcp

CaF₂

Hex.

YH_{0-0.23}

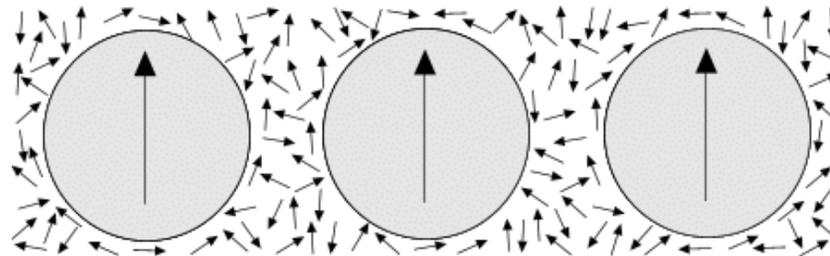
YH_{1.8-2.1}

YH_{2.75-3.0}

Spin-spin exchange interaction between atoms mediated by RKKY (Ruderman-Kittel-Kasuya-Yoshida) interaction spin polarization of conduction electrons

$$\mathcal{H}_{ex} = - \sum_n \Gamma(\vec{r} - \vec{R}_n) \hat{s}(\vec{r}) \cdot \hat{S}_n$$

-4f metals, 4f intermetallics



Magnetic ordering temperatures of several ferromagnets

Material	T_c (K)
Co	1403
Fe	1043
Ni	631
Gd	293
GdNi ₂	79
GdNi	64
GdCl ₃	2.2
SmCo ₅	993
Nd ₂ Fe ₁₄ B	585

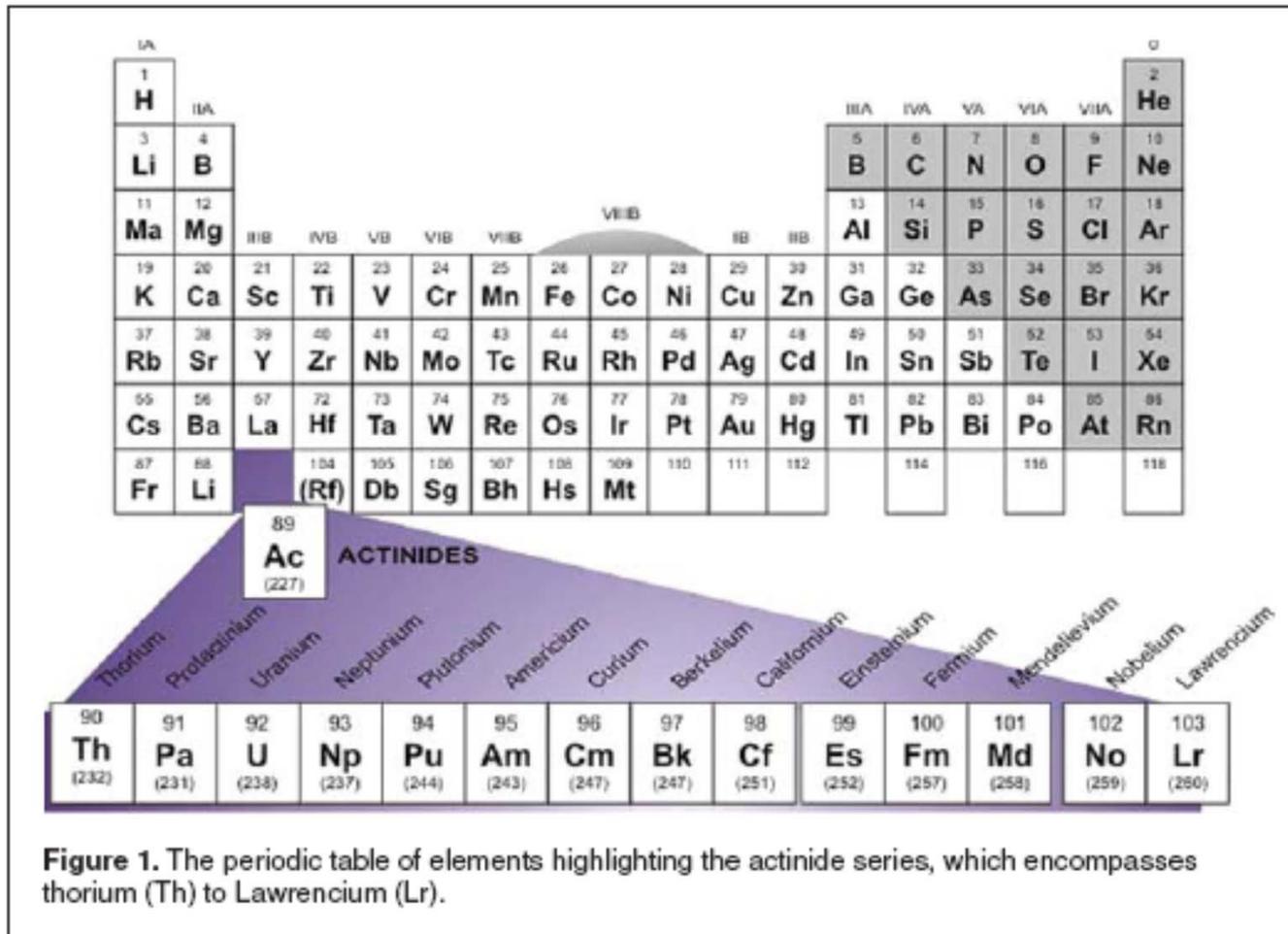
***d*-metals: small magnetic moments
but strong interaction – *3d* band**

***4f*-metals: large moments from
Hund rules, but *4f* states deep
inside atoms – how they interact?**

GdH₂ – 21 K

GdH₃ – 3.3 K

But we are interested in....**The Actinides**



Gradual filling of 5*f* states.....6*d* states remain with low occupancy...also 7*s*

How the 5*f* states look like?

Actinides – 5*f* states

2 distinctly different parts of the series

Lanthanides

6*d*,7*s*

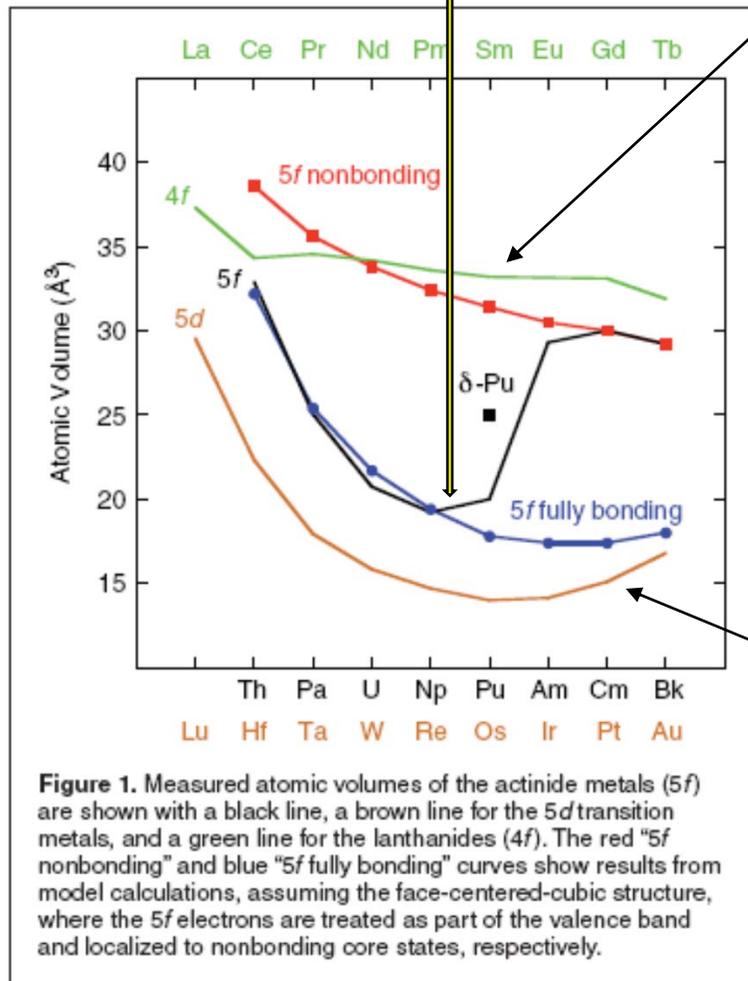
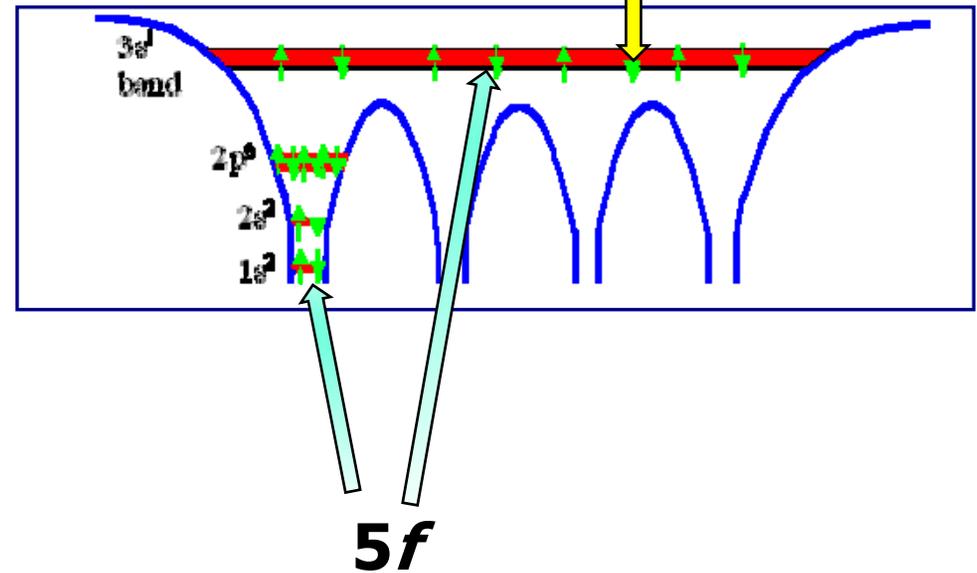
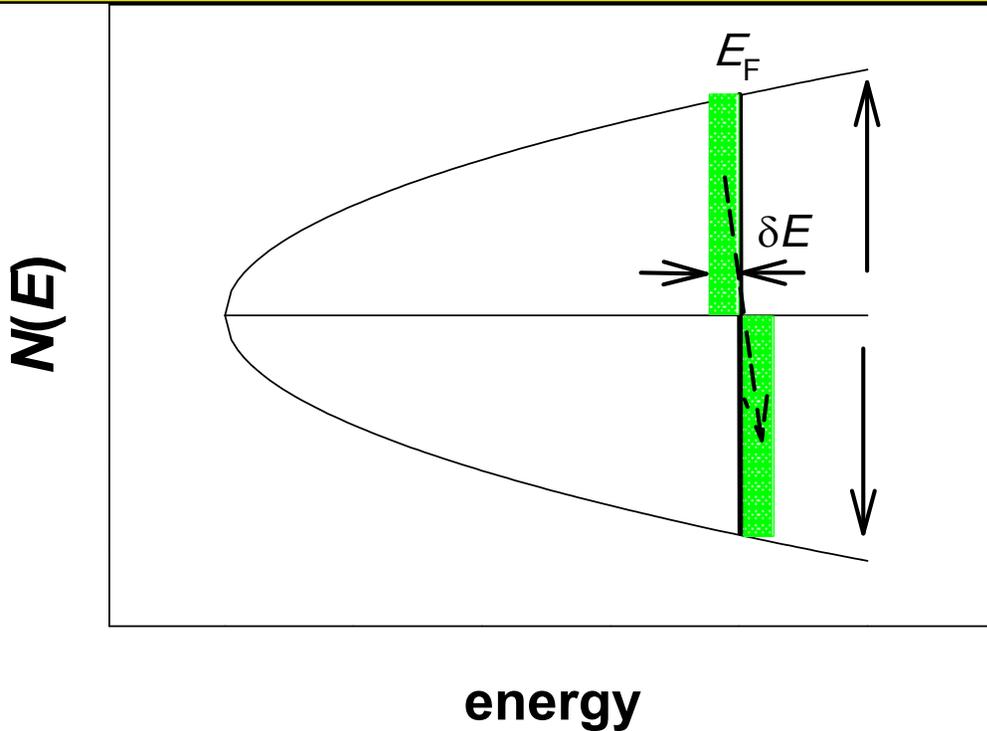


Figure 1. Measured atomic volumes of the actinide metals (5*f*) are shown with a black line, a brown line for the 5*d* transition metals, and a green line for the lanthanides (4*f*). The red "5*f* nonbonding" and blue "5*f* fully bonding" curves show results from model calculations, assuming the face-centered-cubic structure, where the 5*f* electrons are treated as part of the valence band and localized to nonbonding core states, respectively.



Transition metals

Magnetic moments and their order can arise also in band case – exchange interactions cause unequal occupancy of states with different spin



Sub-bands with opposite spins

Exchange interactions – repulsion of electrons with the same spin is lower than for opposite spins

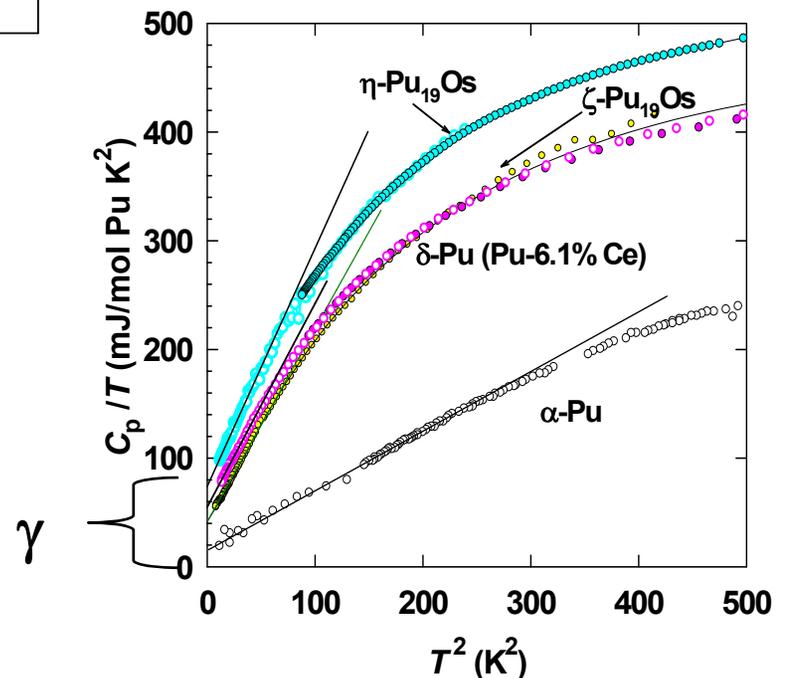
$U * N(E_F) > 1$ spontaneous splitting.....

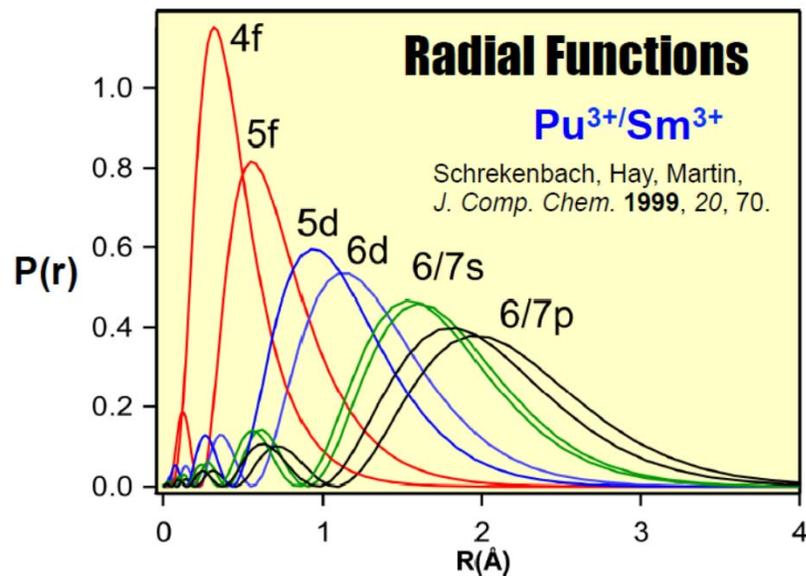
Stoner criterion

$N(E_F)$ can be measured!!!

$$C = \gamma T + \beta T^3; \quad \gamma = 1/3 k_B \pi^2 N(E_F);$$

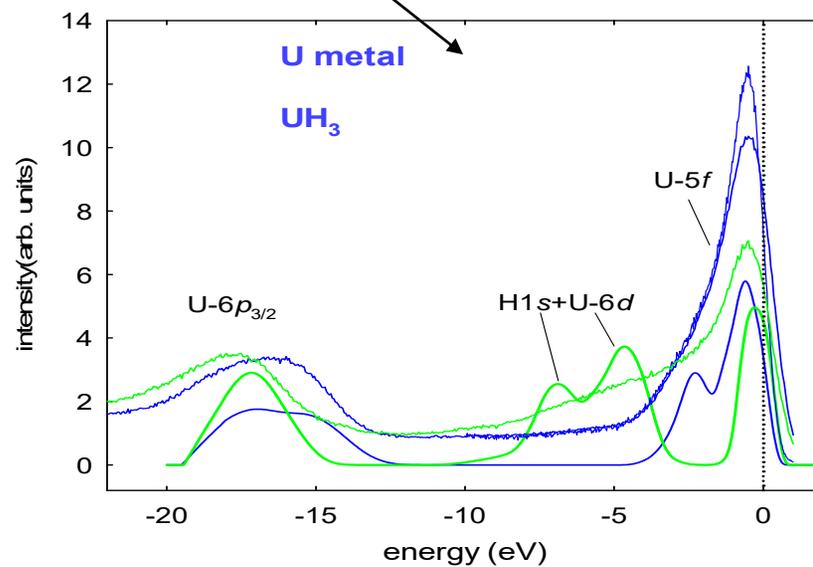
$$\beta = (1944 * 10^3) n / \Theta_D^3$$





Polar character of bonding –H likes to suck electrons from electropositive elements

Photoelectron spectroscopy



Electronic properties of α -UH₃ stabilized by Zr

I. Tkach,¹ M. Paukov,¹ D. Drozdenko,¹ M. Cieslar,¹ B. Vondráčková,¹ Z. Matěj,¹ D. Kriegner,¹ A.V. Andreev,²
 N.-T. H. Kim-Ngan,³ I. Turek,¹ M. Diviš,¹ and L. Havela¹

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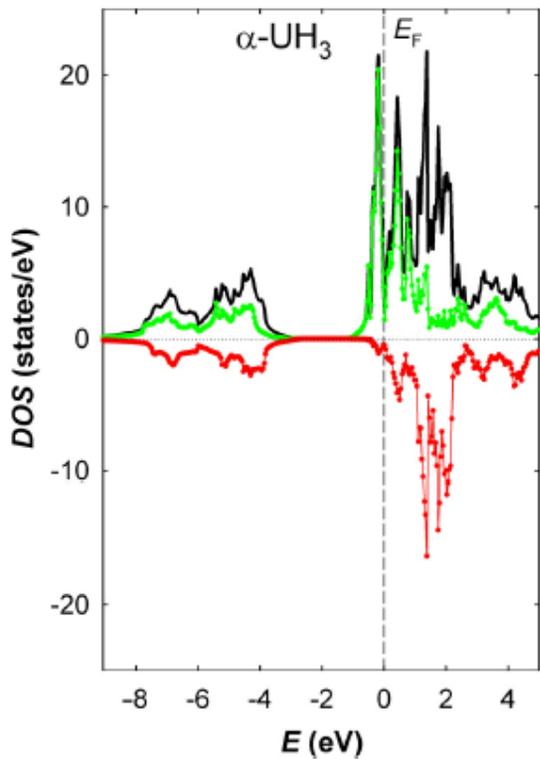


FIG. 10. (Color online) Total (black) and spin-resolved (green and red) density of states for α -UH₃ calculated using fully relativistic FPLO method.

TABLE II. Site projected occupancies of individual states for α -UH₃ (at experimental lattice parameter) and bcc U with the same lattice parameter. Occupancies of particular states in interstitial areas are not included.

State	α -UH ₃	bcc U
U-6d	0.63	1.22
U-7s	0.03	0.42
U-5f	2.71	2.62
H-1s	1.74	

Charge transfer to H?
6d, 7s yes
5f no

Robust ferromagnetism
Suppression of 5f-6d hybridization??

Increase of atomic radius between H and H⁻

H : 0.032 nm
H⁻ : \approx 0.1 nm

superconductivity

Hill limit

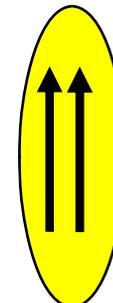
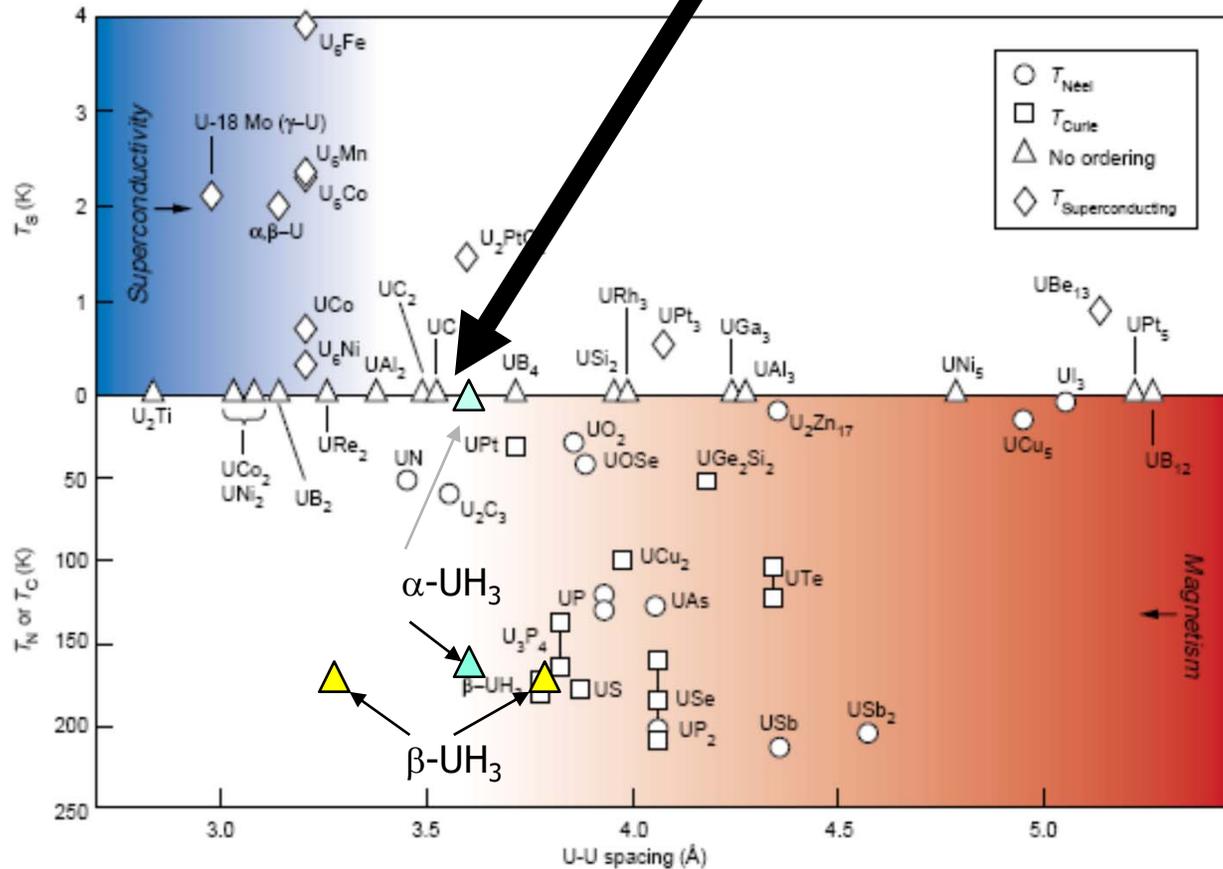
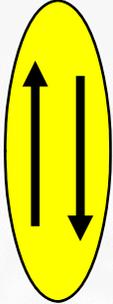


Figure 24. Hill Plot for Uranium Compounds

The Hill plot shows the superconducting or magnetic transition temperatures vs interatomic spacing separating the f electron atoms. We augmented the original Hill plot for uranium compounds to include more data, in particular, the transition temperatures of the

Magnetism – atoms cannot be too close together

f-electron spacings (blue quadrant), and the magnetic compounds at large f-electron spacings. The heavy-fermion superconducting compounds are exceptions. Although the spacing between the uranium atoms in those compounds is fairly large, the f electrons are still not fully localized and can condense into a superconducting state.

H₂ attacks U breaking it into fine pyrophoric powder (detrimental for devices)

**Can be H₂ used to tune Properties of U?
Yes!!!**

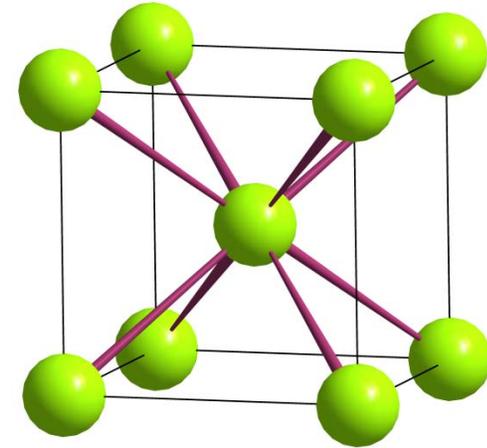
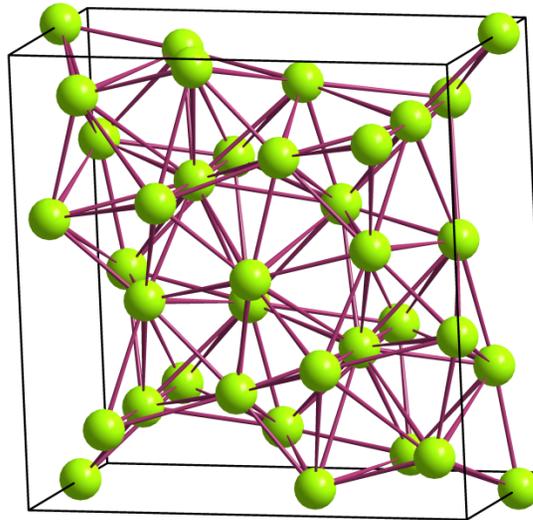
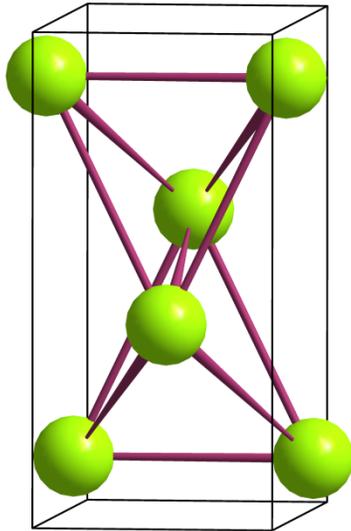
**Can be U used to store H₂?
Yes...works very well
For Tritium**

**U forms UH₃. Pressure of H₂ at 700 K – 1000 mbar
at 300 K – 10⁻⁴ mbar**



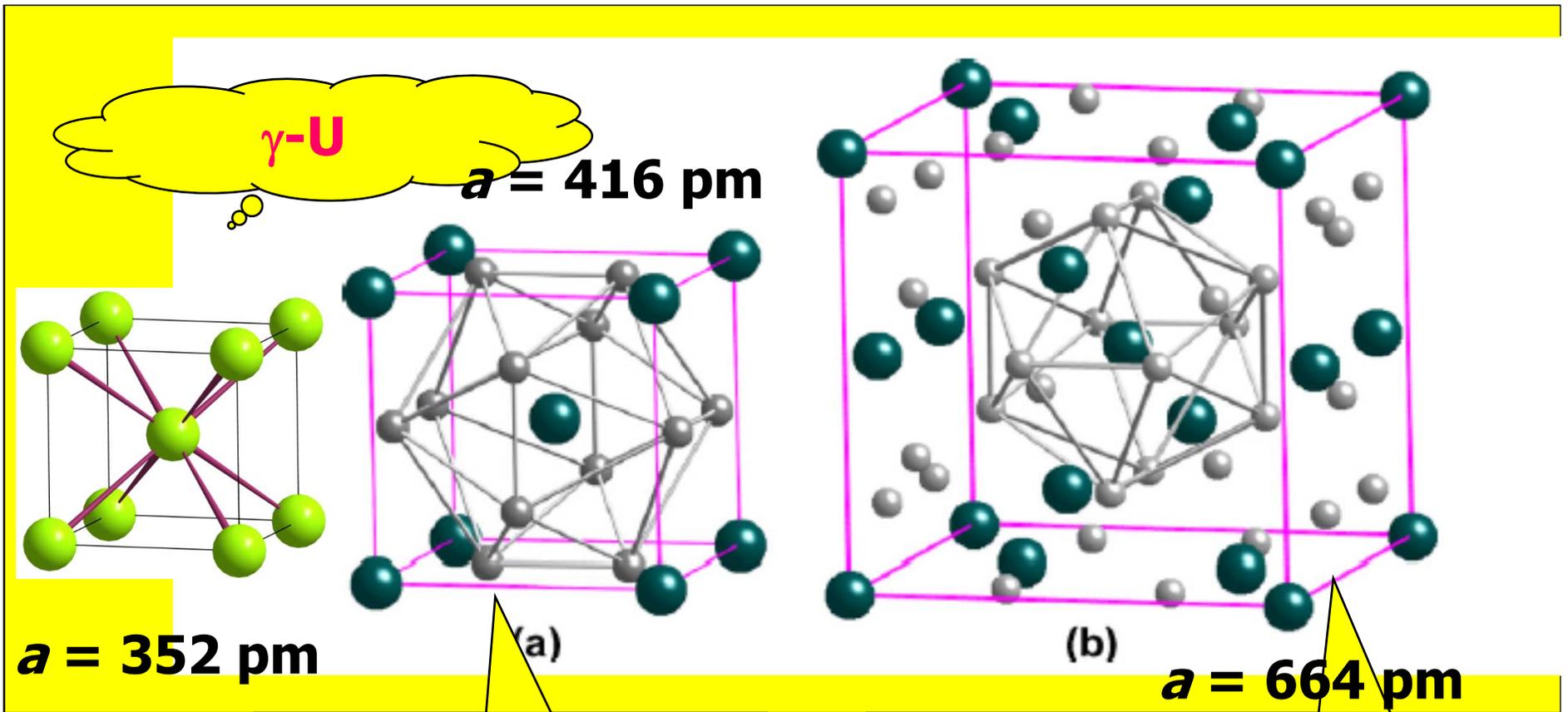
Uranium – 3 allotropic phases

Can be stabilized by > 18 at.% Mo
how about less Mo
and ultrafast cooling?



	α	β	γ
structure	orthorhombic	tetragonal	cubic (<i>bcc</i>)
density	19.07 g/cm ³	18.17 g/cm ³	17.94 g/cm ³
shortest U-U distance	2.837 Å	2.889 Å	3.067 Å
temperature	below 942 K	942-1049 K	1049-1408 K

γ -U stable with respect to irradiation, corrosion, hydrogen.. Low enriched U nuclear fuel
But - what are low-*T* electronic properties ??



α -UH₃
 $d_{\text{U-U}} = 360 \text{ pm}$

β -UH₃
 $d_{\text{U-U}} = 330 \text{ pm}$

Ferro- or para ? Either $T_c = 165 - 170$
 or 0
Lawson et al. 1991

Ferro- $T_c = 165 - 170 \text{ K}$, $\mu = 0.9 \mu_B$
Trzebiatowski et al. 1952

superconductivity

Hill limit

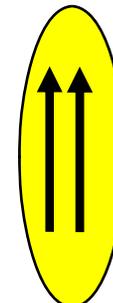
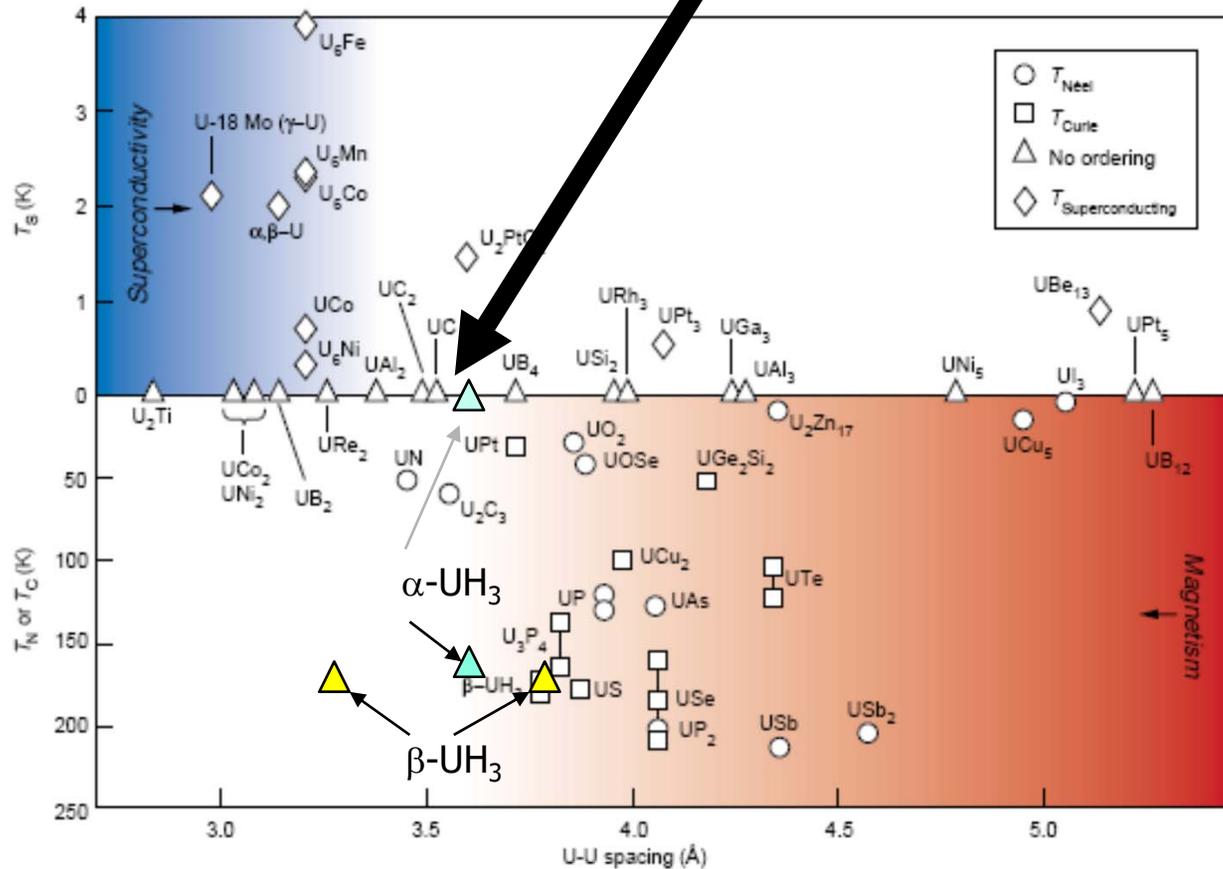
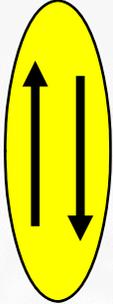


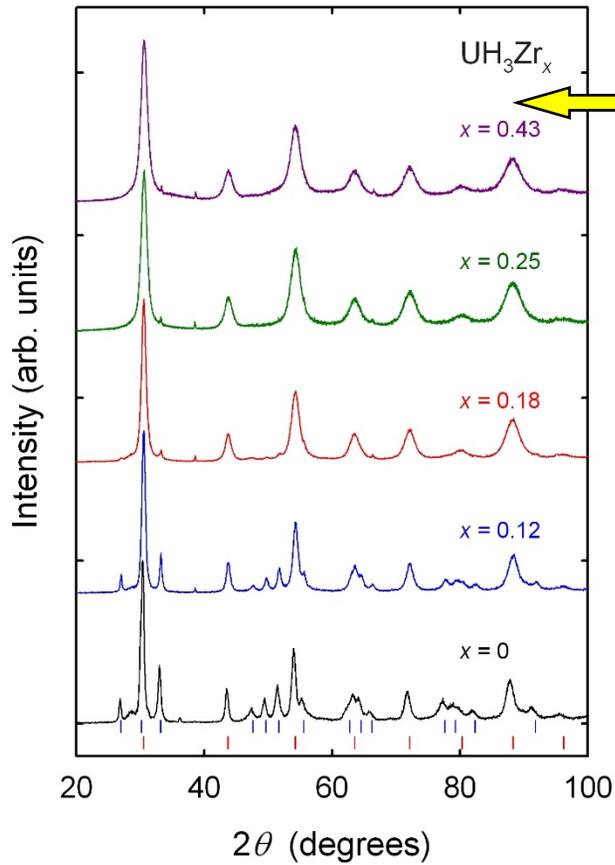
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α-UH₃
D = 10-20 nm

Magnetic Properties of α-Uranium Hydride
 by
 A. ŚLIWA and W. TRZEBIATOWSKI
 Presented by W. TRZEBIATOWSKI on March 10, 1962

BULLETIN DE L'ACADÉMIE
 POLONAISE DES SCIENCES
 Série des sciences chimiques
 Volume X, No. 5, 1962

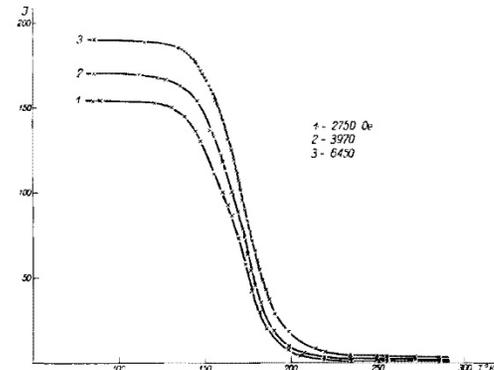
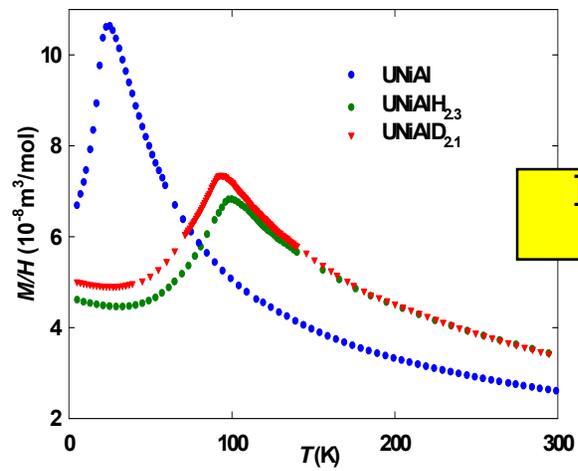


Fig. 1. Magnetization of α-UH₃ containing 25% β-UH₃

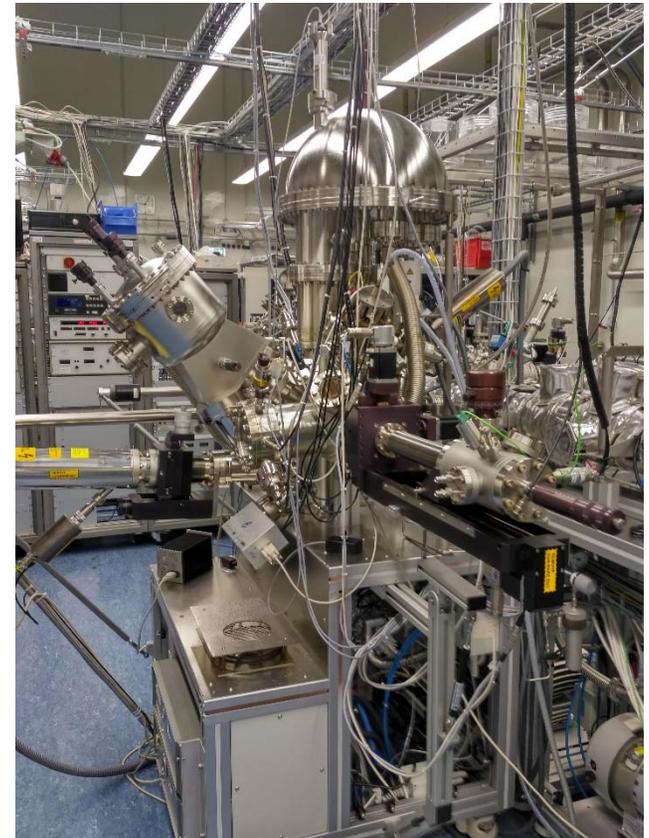


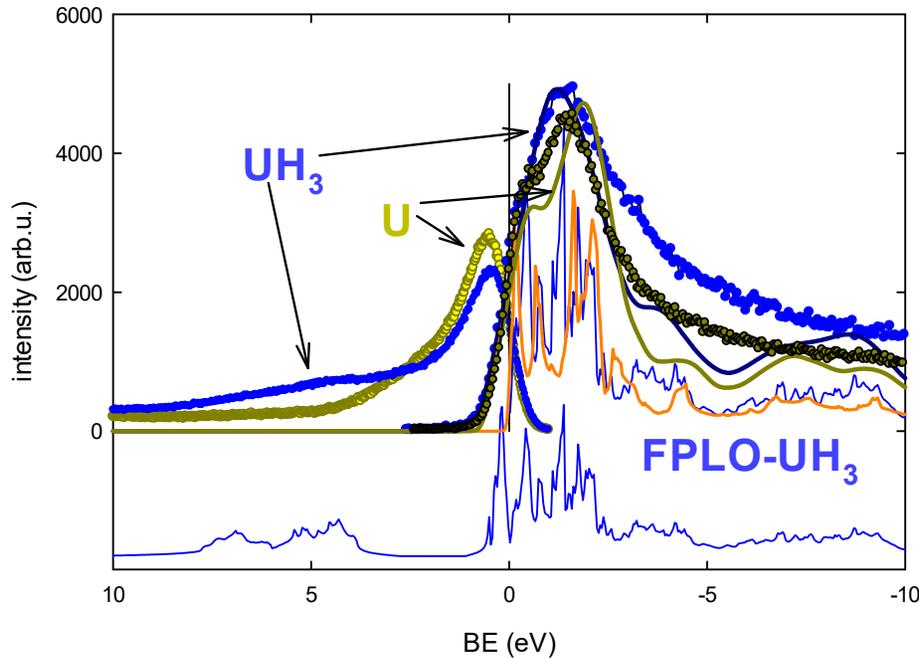
Increase of ordering T
 in antiferromagnet

**UH₃ –
 ferromagnet
 with T_C = 165
 K**

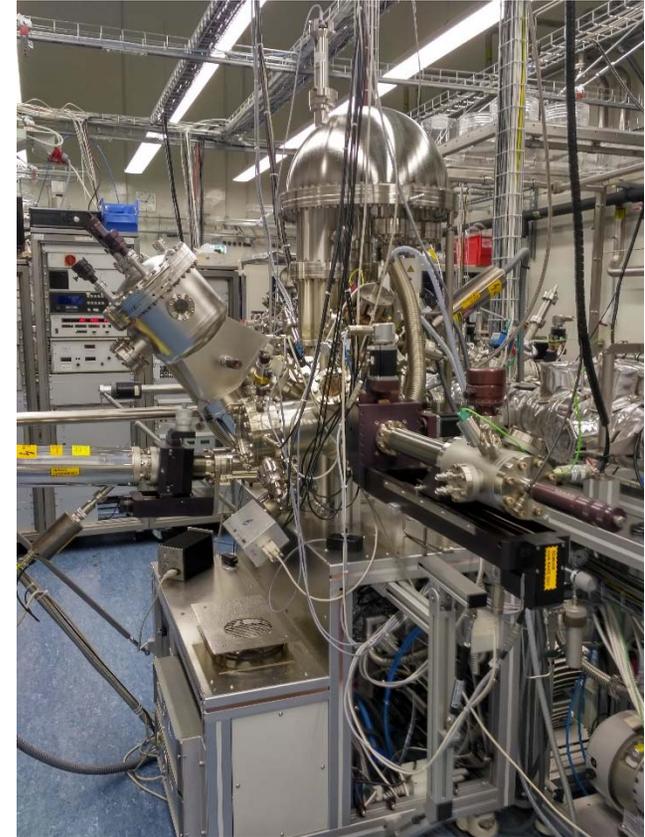
Synthesis of films by sputter deposition

- U and Mo targets with voltage independently controlled
- Thoriated W heated filament for stabilizing plasma
- Ar gas (10^{-3} mbar) , variable concentration of H₂
- Si wafer or fused silica substrates
- Purity monitored by O-1s (XPS) and O-2p (UPS) lines





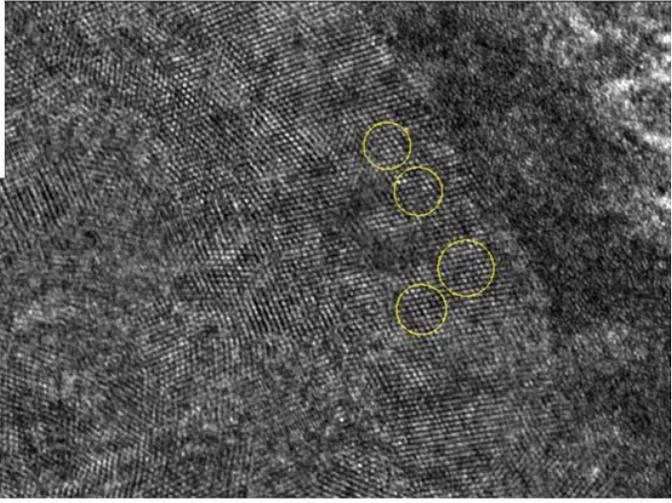
Combined with BIS



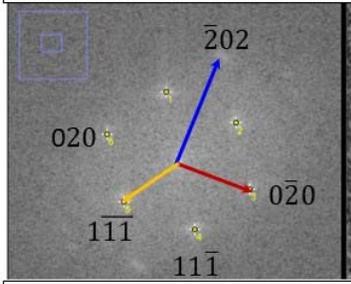
BIS shows the s-o splitting for U metal. It is smeared in UH_3 due to spin-up and spin-down splitting. More spectral weight at higher energies... $6d$ states.

	Area	Mean	Min	Max	X	Y	R
1	0.000	190	190	190	9.671	8.663	0.304
2	0.000	170	170	170	10.953	9.222	0.311
3	0.000	149	149	149	11.198	10.449	0.277
4	0.000	197	197	197	10.191	11.172	0.296
5	0.000	170	170	170	8.886	10.682	0.297
6	0.000	146	146	146	8.589	9.428	0.272

Interplane distances from FFT

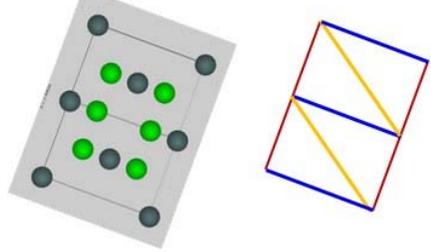


FFT, diffraction planes

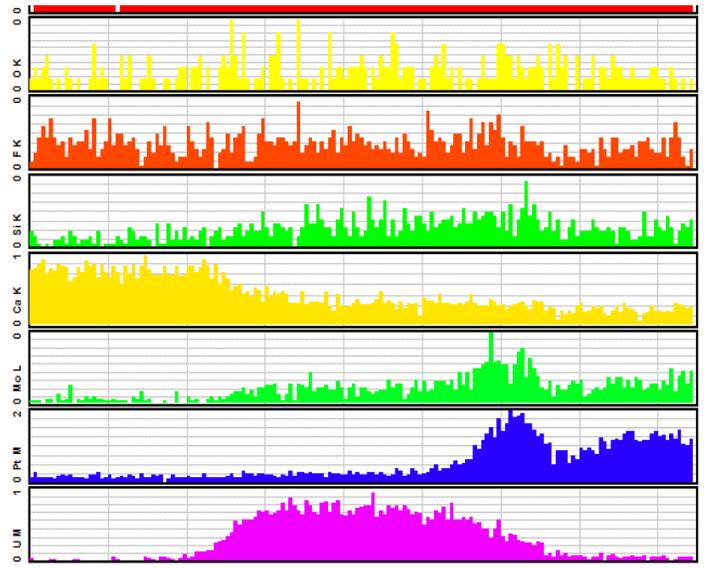


Orientation of cristal in real espace

Axis [011] of UH₂ $d_{111}=0.302/d_{200}=0.2745$



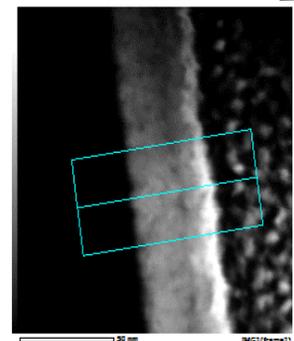
100 diffracting planes
110 diffracting planes
111 diffracting planes



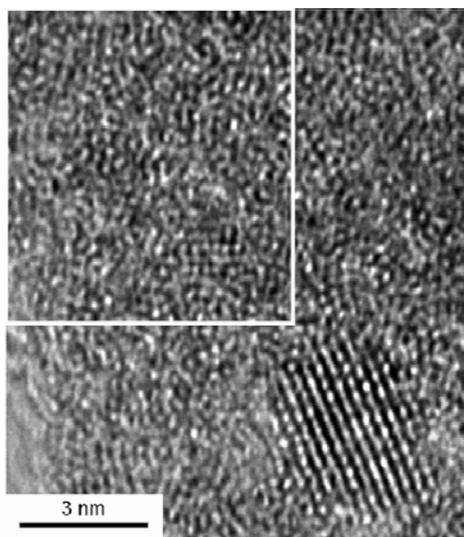
- OK
- FK
- SiK
- CaK
- MoL
- PtM
- UM

UH₂ film

Pt

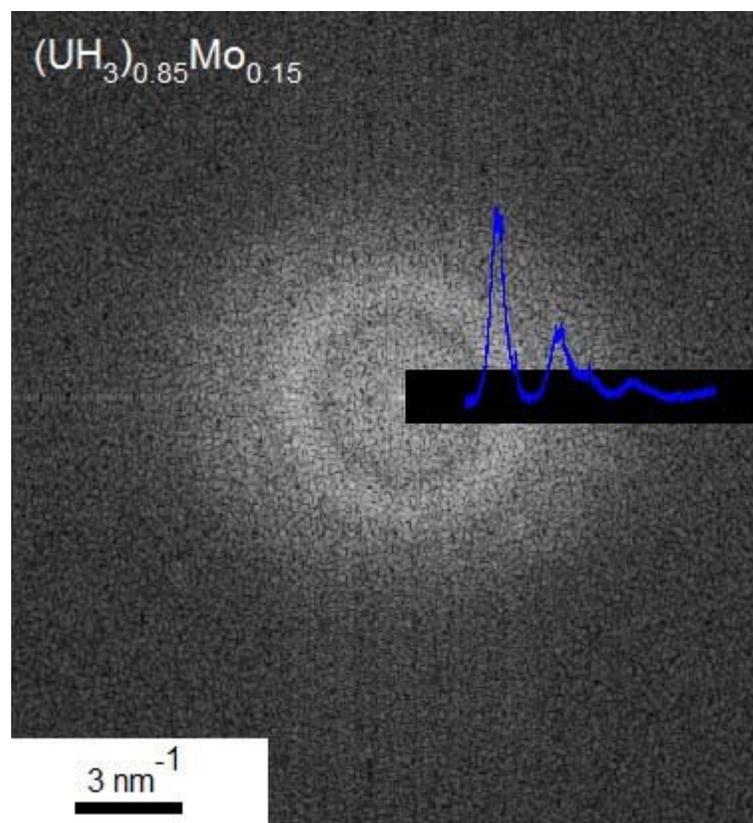


CaF₂

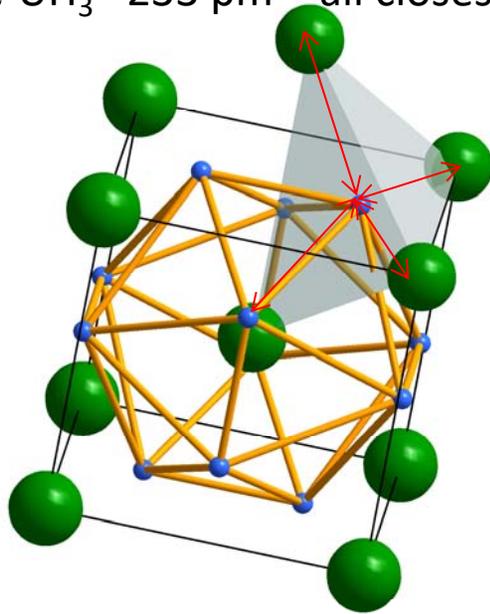


Ultrananocrystallinity with high anisotropy means a non-collinear ferromagnetism with random distribution of easy magnetization directions

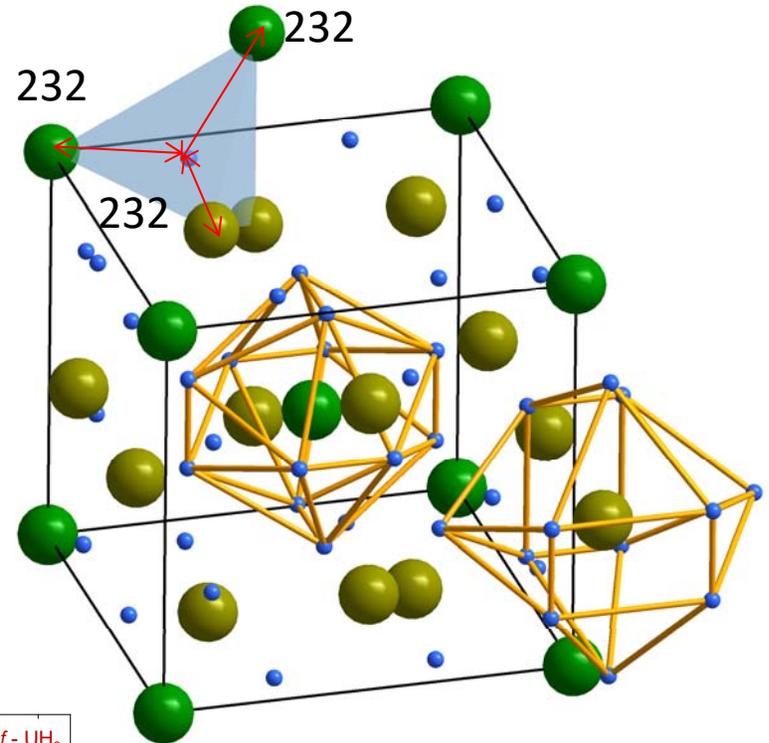
HRTEM



α -UH₃ 233 pm – all closest U atoms

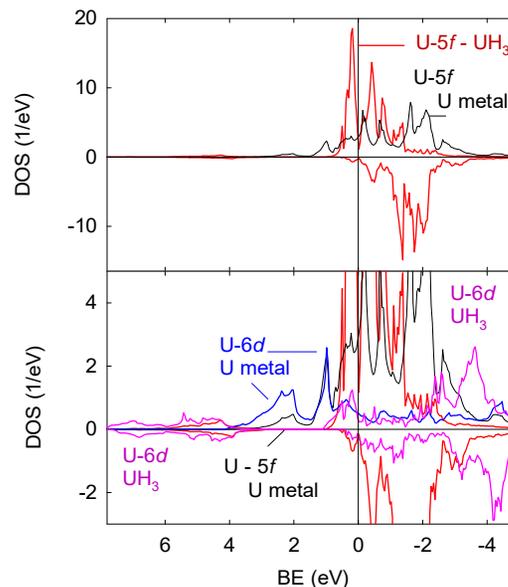
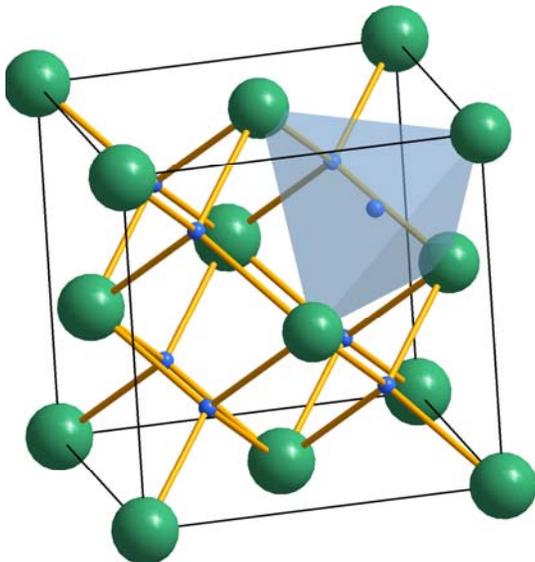


β -UH₃



**Identical
tetrahedra
..as molecules..
which
determine T_C**

UH₂ 232 pm – all closest U atoms

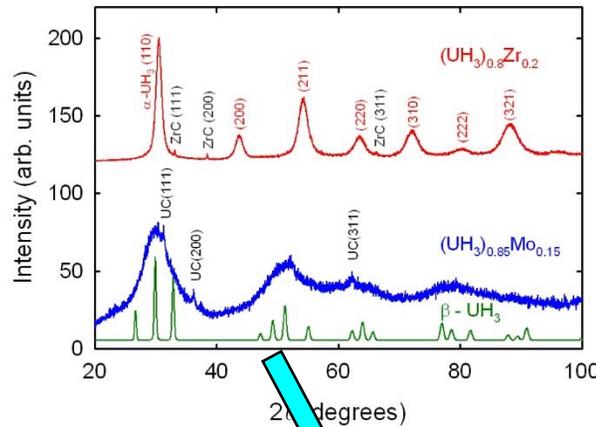


**$r_U = 156$ pm
 $r_H = 37$ pm**

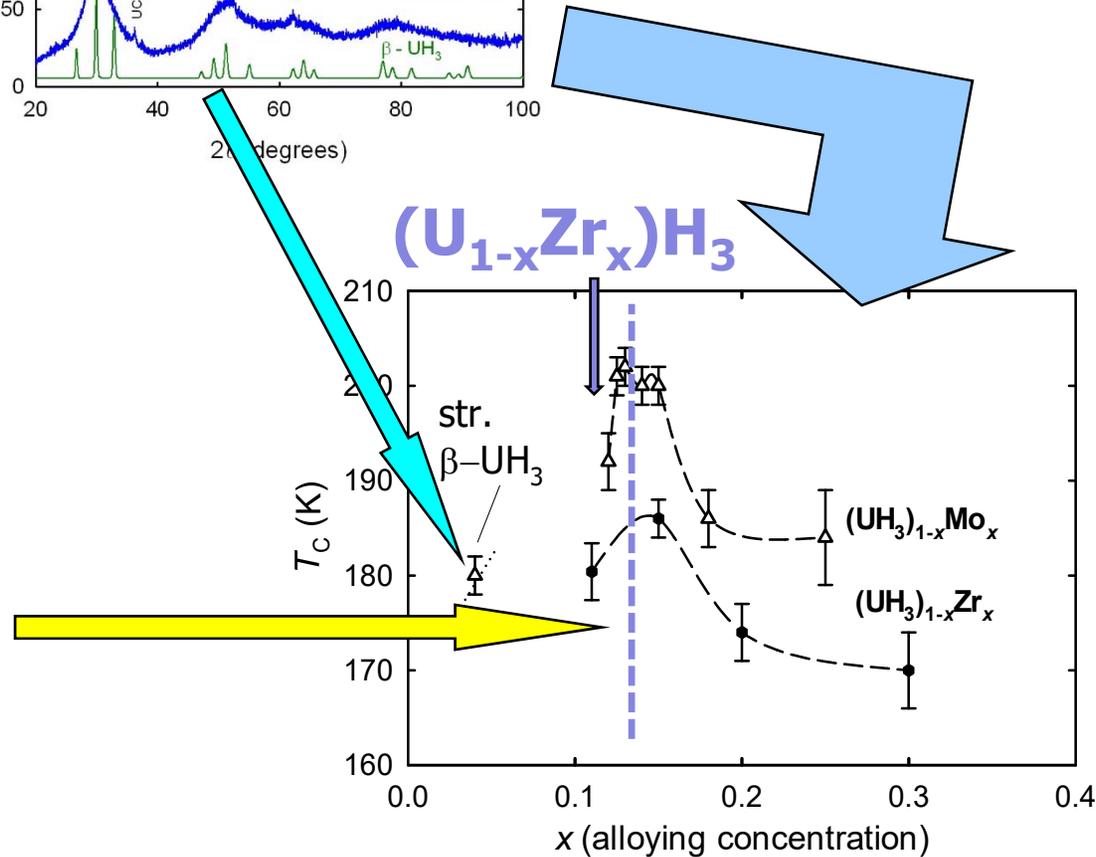
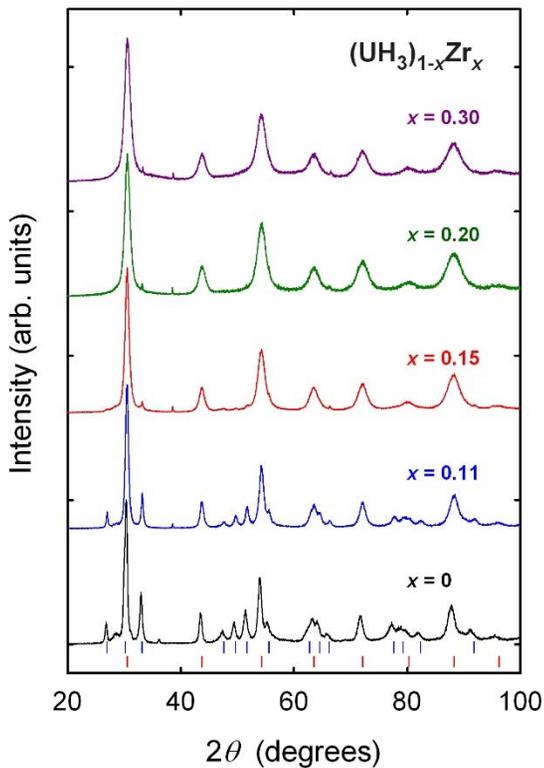
**$r_H = 76$ pm in UH₃
or UH₂**

On the way to H⁻
Polar bonding in metallic state

Doping by Mo or Zr



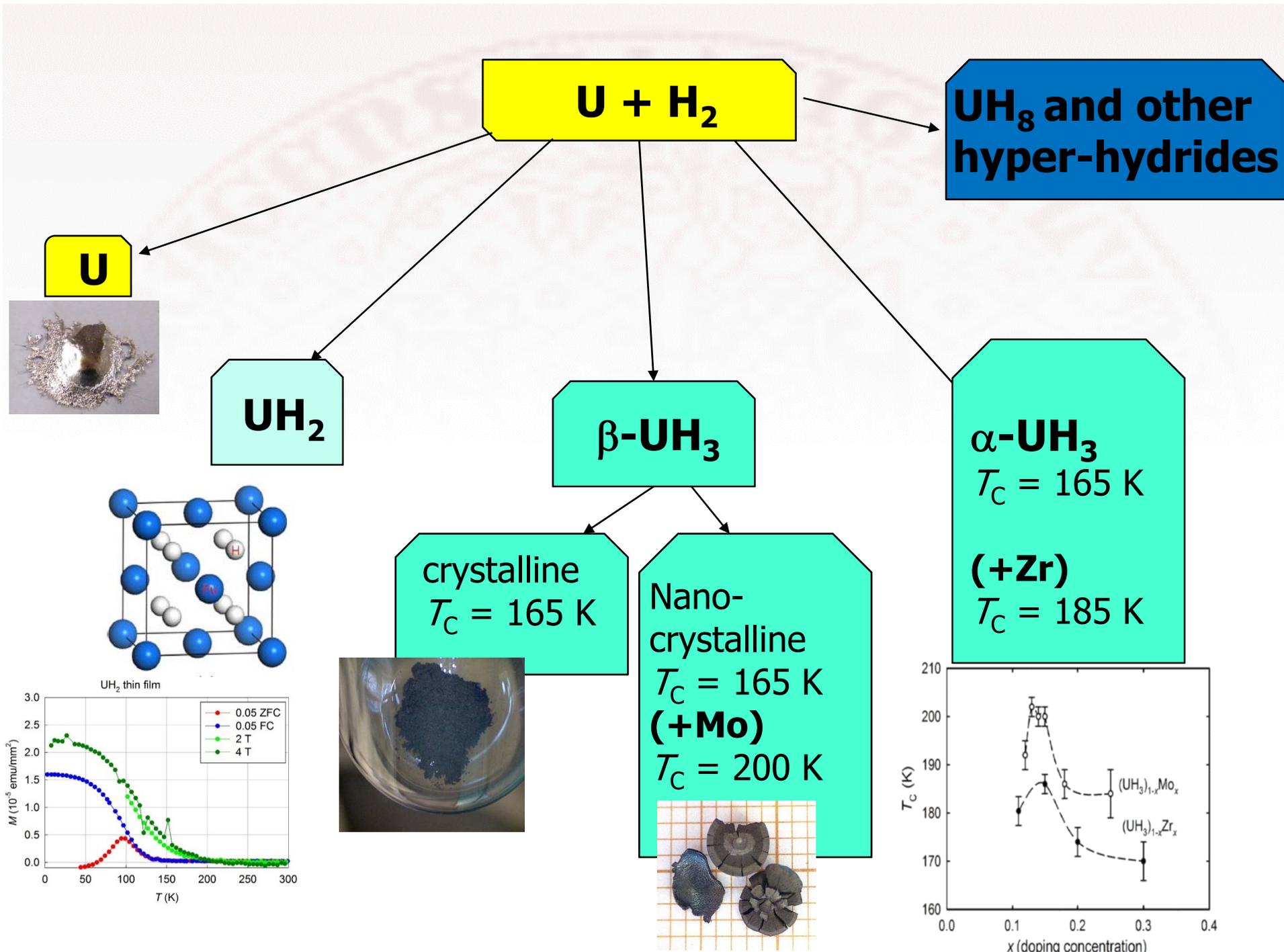
$\beta\text{-UH}_3$
 $D = 2\text{-}3\text{ nm}$



$\alpha\text{-UH}_3$

$D = 10\text{-}20\text{ nm}$

**5f band
ferromagnets?**



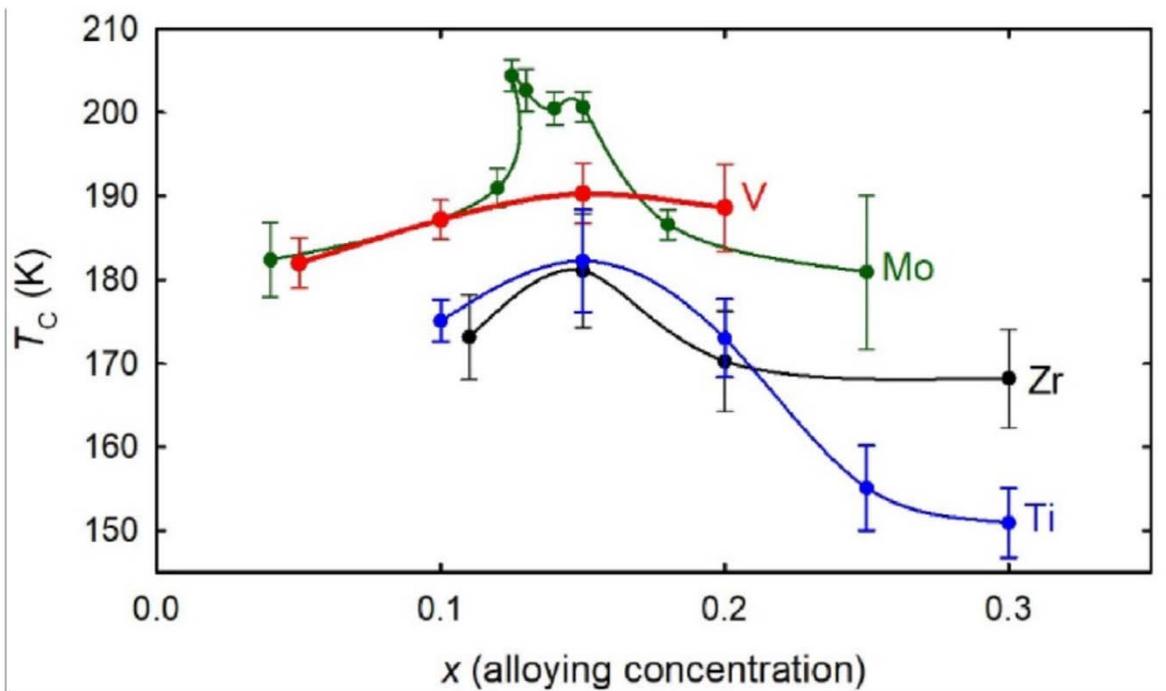


Fig. 6. Concentration dependence of the Curie temperature in the series $(\text{UH}_3)_{1-x}(\text{V-Mo-Zr-Ti})_x$.

Maximum T_c values found universally for the T concentrations 12-15 at.% suggest that 1/8 (i.e. 0.125) of the U atoms can be safely replaced still maintaining H atoms stable inside the tetrahedra. **H-U ratio increases. U electropositive.**

U-H interaction plays a dominant role!



Conclusions:

In U-H there is evidence for a charge transfer towards H, similar to e.g. pnictides.

In U-H, partial depletion of $6d$ and $7s$ states ($5f$ band remains at E_F) may lead to strong magnetism, normally negatively affected by the $5f - 6d$ hybridization. Strong magnetism and partial localization despite short U-U spacings

Robust ferromagnetism opens an avenue for thin-film devices (Giant magnetoresistance, large exchange bias)

Old publications can be seen from a new perspective....combined physics-chemistry view brings benefits

Main collaborators: Poland (prof. Kim et al.)

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Volodymyr Buturlim – bulk properties

Ilja Turek, Martin Divis, Jindra Kolorenč, Dominik Legut - calculations

Daria Drozdenko, Zdenek Matej, Milan Dopita, Mirek Cieslar, Mayerling

Martinez – XRD, TEM, EBSD, EXAFS, PDF

Sasha Koloskova, Frank Huber, Thomas Gouder, A. Seibert – XPS, UPS, BIS

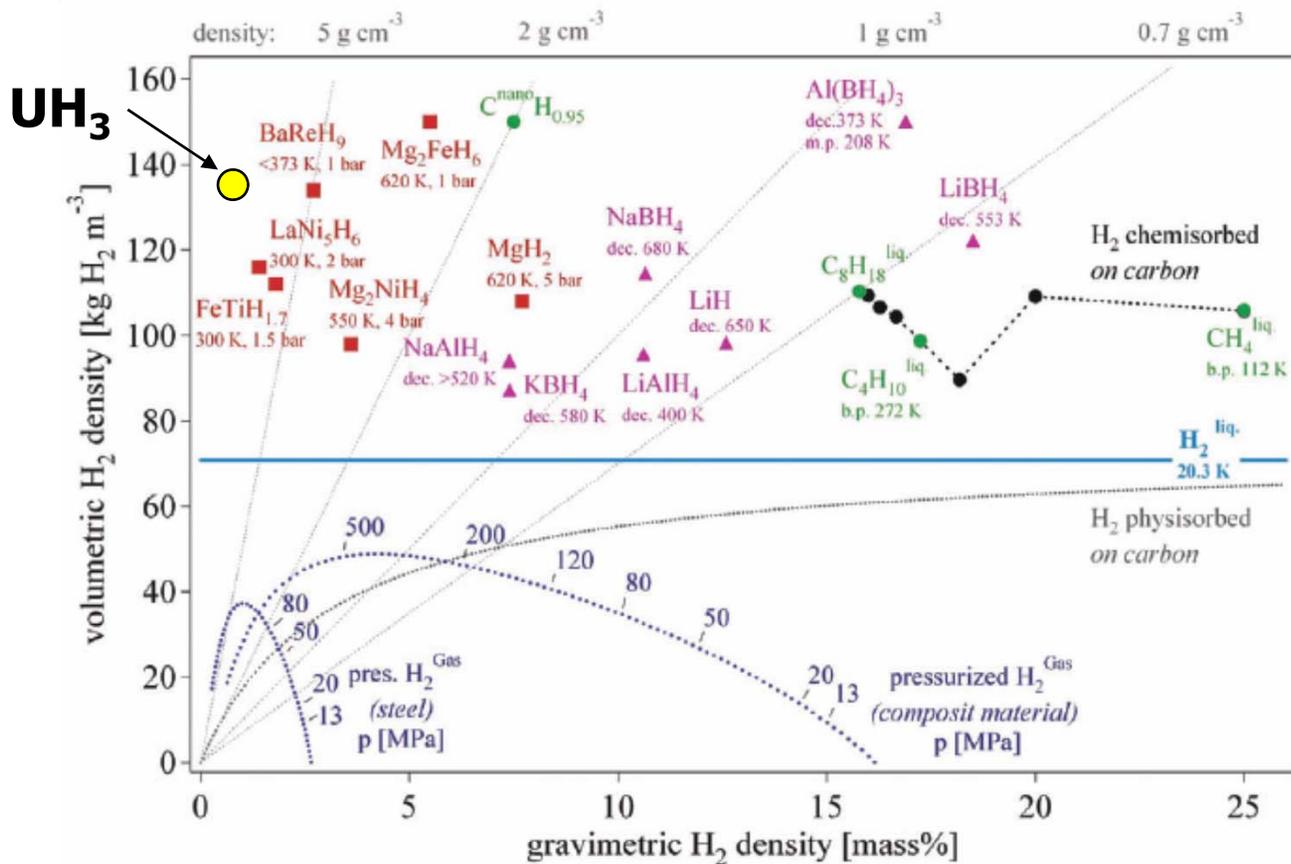


Fig. 7 Volumetric and gravimetric hydrogen density of some selected hydrides. Mg_2FeH_6 shows the highest known volumetric hydrogen density of $150 \text{ kg}\cdot\text{m}^{-3}$, which is more than double that of liquid hydrogen. $BaReH_9$ has the largest H/M ratio of 4.5, i.e. 4.5 hydrogen atoms per metal atom. $LiBH_4$ exhibits the highest gravimetric hydrogen density of 18 mass%. Pressurized gas storage is shown for steel (tensile strength $\sigma_v = 460 \text{ MPa}$, density $6500 \text{ kg}\cdot\text{m}^{-3}$) and a hypothetical composite material ($\sigma_v = 1500 \text{ MPa}$, density $3000 \text{ kg}\cdot\text{m}^{-3}$).